## Multiscale Modeling and Simulation of Advanced Photovoltaic Devices

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# Outline

- High efficiency photovoltaics and efficiency limits
- Multiscale modeling of c-Si/a-Si:H heterojunction solar cells
- Multiscale modeling of Si tandem solar cells
- Multiexciton generation in nanowires for advanced concept solar cells

# **Photovoltaics (PV)**



First Generation single crystal Si PV technology

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# **Silicon Heterojunction Solar Cells**



- > Carrier selective contact structures are used to obtain high efficiencies
- The intrinsic hydrogenated amorphous silicon (a-Si:H(i)) is crucial as it provides passivation and carrier selectivity
- If the a-Si:H(i) layer is too thick, photocurrent suppression occurs
- Understanding defect transport through a-Si:H(i) is necessary to optimize the layer thickness and quality



# First Principles Study of c-Si/a-Si:H Interface



Simulated c-Si/a-Si:H interface using LAMMPS/DFT



<sup>\*</sup>R. Vatan, D. Vasileska, S. M. Goodnick, J. Phys. Chemistry (in review)



# Transport Through a-Si:H(i)



- Traditional methods do not model non-local transport well

Carlo

10-6

Time Scale (sec)

10-3

10<sup>0</sup>

Ensemble

Monte Carlo

10-12

10-9

10<sup>-9</sup>

10-1

10-15

## **Transport Through KMC Domain**

#### Thermionic Emission







## **Injection and Extraction**

#### How do carriers get into the a-Si:H(i) layer? Multi-phonon injection



Injection depends on :

- ✓ Defect density
- ✓ Optical phonon energy
- ✓ Electric field in the a-Si:H(i)

How do carriers get out of the a-Si:H(i) layer? Poole-Frenkel and Defect Emission



Extraction depends on : ✓ a-Si:H(i) thickness ✓ Electric field in the a-Si:H(i)





For Ohmic transport Injection time = extraction time



- Current Density = Charge/Time
- For a surface density ~ 10<sup>12</sup> cm<sup>-2</sup> at the heterointerface, transit time required for ~ 40 mA/cm<sup>2</sup> is 40 µsec (Ohmic transport)
- > The KMC predicts transit time  $10^{-5} \rightarrow 10^{-1}$  sec
- ➢ Higher transit times → lower currents, higher surface charge density → higher currents
- KMC simulations indicates that the extraction supports lower currents, i.e. photocurrent suppression



# **Two Junction Si Tandems**

 Detailed balance calculation of the two-junction series connected efficiency



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# **Epitaxial Combinations for Si**

#### Bandgap too high

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Large lattice mismatch



- GaP is the only lower bandgap III-V which is nearly lattice-matched to Si
- Perovskite/Si tandems currently hold record (~27%)

### **Dilute GaNP Nitrides**

Large bandgap reduction through incorporation of a small fraction of N in III-Vs



## **Tight Binding Modeling of Dilute Nitrides**

- Excellent agreement with experimental bandgap versus N mole fraction data
- Optical absorption calculated from direct allowed transition in dipole approximation



### **Transport Simulation**

- The TB basis was used in full band cellular Monte Carlo simulation of the mobility of electrons and holes in binary GaP and alloys of GaNP
- Polar optical, acoustic, deformation potential optical, ionized impurity including compensation



## **TCAD Device Simulation**

- Silvaco ATLAS simulation of 3-terminal device using simulated optical absorption and transport data
- Lifetime of the GaNP layer critical parameter in determining the maximum conversion efficiency



## **Multi-Exciton Generation Solar Cells**

Ultrafast energy relaxation processes a major loss mechanism in photovoltaic devices

Impact ionization or multiexciton generation uses the excess photon energy to increase the quantum yield above 100%





### **MEG in Nanowires**



Figure 4. The exciton quantum yield as a function of relative photon energy  $(h \nu E_g)$  for PbSe nanorods and nanocrystals. The energies of the first excitonic absorption are listed for each batch of nanostructures. Lines are best fits to equation 2. The MEG efficiency is 0.78 for nanorods and 0.41 for nanocrystals. Literature values taken from Beard, et al. [27].

P. D. Cunningham et al., Proc. SPIE Vol. 8256, 2012,

17

#### Band Structure of Hexagonal Cladded Nanowires along 111

- Empirical TB sp<sup>3</sup>d<sup>5</sup>s\* including spinorbit interaction. Parameters taken from Purdue group.\*\*
- The periodicity and the supercell of the nanowire is created using the primitive basis vectors of the material and the nanowire cross-section dimensions.



\*\*TB Boykin, G Klimeck, F Oyafuso , Physical Review B 69 (11), 115201





### **Scattering Rates**

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Fermi's Golden Rule is used to calculate all scattering rates:

$$W(k_{\mu},k_{\nu},\mathbf{q}) = \frac{2\pi}{\hbar} \left| \left\langle \psi_{k_{\nu},q} \left| \delta H \right| \psi_{k_{\mu},q} \right\rangle \right|^2 \delta \left( E_{k_{\nu}} - E_{k_{\mu}} \pm \hbar \omega_q \right)$$

Bulk-like phonons are assumed in the cases of def. potential and polar optical phonon scattering.



#### Cellular Monte Carlo Simulation of Energy Relaxation in Nanowires

The plots below show the relaxation of electrons after the initial excitation at t = 0 ps.



2nm x 2nm InAs Carrier Relaxation

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3nm x 3nm InAs Carrier Relaxation

Electrons' energy is initialized to a Gaussian distribution with a half-width of 100 meV at t = 0.

Hathwar et al., J. Phys. D 52 (2019)

#### Carrier Generation due to Electron-Photon Interaction





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> The excess energy of electrons in the CB is significantly reduced due to partitioning between VB and CB.

## MEG for 2 nm InGaAs with 1 nm InP



Linear behavior in MEG excitation curve due to partitioning of photon energy between VB and CB

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